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Hydrogen-Atom Direct-Entry Mechanism into Metal Membranes

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ABSTRACT

The hydrogen-atom direct-entry mechanism is used to explain why the steady-state hydrogen permeation current density through a metal membrane is directly proportional to the cathodic current density, i_c , and is independent of the membrane thickness when i_c is small.

Bagotskaya¹ and later $Frumkin²$ postulated that hydrogen atoms enter directly into a metal membrane and do not go through an intermediate adsorbed state. The mechanism they proposed is

$$
H^* + M + e^- \underset{k_{-1}}{\overset{k_1}{\rightleftharpoons}} MH_{abs} \tag{1}
$$

$$
H^* + M + e^- \stackrel{k_2}{\rightarrow} \text{MH}_{\text{ads}} \stackrel{k_3}{\rightarrow} \frac{1}{2} H_2 + M \tag{2}
$$

where k_1, k_{-1}, k_2 , and k_3 are rate constants.

Bockris *et al. 3* checked this mechanism by considering the following equations

$$
J_{\infty} = F\left[k_1\left(1 - \frac{C_o}{C_s}\right)e^{-\alpha_1 a \eta} - k_{-1}C_o e^{(1-\alpha_1) a \eta}\right]
$$
 [3]

$$
i_c = i_o e^{-\alpha_2 a \eta} \tag{4}
$$

$$
j_{\infty} = \frac{DF}{L} C_{\infty}
$$
 [5]

where j_{∞} is the steady-state permeation current density, η is the overpotential, $a = F/RT$, C_0 is the hydrogen concentration directly beneath the cathode surface, C_s is the saturation value of C_o , α_1 is the transfer coefficient of reaction 1, α_2 is the transfer coefficient of reaction 2, i_c is the cathodic current density, D is the hydrogen diffusion coefficient, and L is the membrane thickness. Bockris *et aI. 3* used Eq. 3 to 5 to claim that the hydrogen permeation current density is proportional to the cathodic current density squared

$$
j_{\infty} \propto i_{\rm c}^2 \tag{6}
$$

Since no experimental evidence exists in literature to support Eq. 6, Bockris et *al.3* concluded that the direct entry mechanism does not predict the observed results and is unable to explain the hydrogen permeation through a membrane.

The hydrogen atom direct mechanism 1,2 is used here to explain why the steady-state hydrogen permeation current density (j_*) is directly proportional to the cathodic current density (i_c) and is independent of the membrane thickness when i_c is small. Assuming that the metal is far from saturation, then $1 - C_o/C_s \approx 1$, and if in reaction 1, k_{-1} is negligible to k_1 , Eq. 3 becomes

$$
j_{\infty} = F k_1 e^{-\alpha_1 a \eta} \tag{7}
$$

The ratio of Eq. 7 to 4 is

$$
\frac{j_{\infty}}{i_{\rm c}} = \frac{Fk_1 e^{-\alpha_1 a \eta}}{i_{\rm o} e^{-\alpha_2 a \eta}}
$$
 [8]

and assuming that the overpotential for the electrode (η) aplies to both reactions 1 and 2

$$
j_{\infty} = i_c \frac{F k_1}{i_o} e^{-(\alpha_2 - \alpha_2)\alpha \eta}
$$
 [9]

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solution in the cell. The permeation rate through a palladium membrane was measured by setting the potential at a fixed value of 0.3 V *vs.* a SCE. These values of the set potentials correspond to a practically zero concentration of absorbed atomic hydrogen on the anodic surface of the mem-

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through palladium membranes is directly proportional to i_c when $i_{\rm e}$ is less than 200 μ A/cm". Heath" reported the same for i_c less than 22 mA/cm . Raczynski and Zacrozymsk *et al.*⁷ demonstrated that $j_x \propto i_c$ for low values of i_c for hydrogen permeation through iron. Also Evseev⁸ reported that $j_{\infty} \propto i_{\text{c}}$ for iron for values of i_{c} between 5 and 80 mA/cm². Note that Eq. 9 shows that j_{∞} does not depend on the thickness of the metal membrane (L) . This is consistent with results presented by Hoare and Schuldiner 9,10 and Bowker and \tilde{P} iercy¹¹ for Pd membranes.

If $\alpha_1 = \alpha_2$, Eq. 9 shows that j_∞ is directly proportional to i_∞ , which is in agreement with previous experiment work \cdot that was done under conditions consistent with the development of Eq. 9. For example, Early⁴ reported that j_x

Experimental

The Devanathan-Stachurski permeation technique¹² was used to investigate the rate of hydrogen permeation through a HY-130 steel and through palladium membranes with an area of approximately 4 cm^2 with thicknesses of 0.15 and 0.025 mm, respectively. The permeation experiments were carried out in a system with two compartments, separated by a bipolar membrane made of HY-130 steel or palladium.

The electrolyte on the cathodic side of the cell for the HY-130 steel membranes was $1M$ Na₂SO₄, 0.4M NaCl, and $1M$ H₃BO₃ and for the palladium membrane was $1N$ H₂SO₄. Atomic hydrogen permeation transients through a HY-130 steel membrane for different applied cathodic potentials were recorded continuously as a function of time. At the beginning of each experiment, the cathodic side of the steel membrane was held at a constant potential of -0.58 V $vs.$ SCE. Next, the potential was stepped to a more negative value after the hydrogen permeation current density reached a steady-state value. After the permeation stabilized at an applied potential of about -0.87 V *vs.* SCE, the potential was switched off, and a decay curve was recorded. The palladium alloy membrane on the cathodic side of the cell was polarized galvanostatically using a low cathodic current density in the range of 59.4 to 594.3 μ A/cm².

The electrolyte on the anodic side of the cell for a HY-130 steel membrane was 0.2M NaOH while for the palladium membrane it was $1N H_2SO_4$. To avoid passivation or dissolution, the anodic side of the HY-130 membrane was electroplated with a thin layer (0.15 to 0.20 μ m) of palladium. The electrodeposition of palladium was carried out in an electrolyte containing $2 \times 10^{-5} M \text{ Na}_2 \text{Pd} (\text{NO}_2)_4$ using a current density of $100 \mu \overline{A}/\text{cm}^2$ for 2 h. After the electrolyte was drained off, the compartment was washed with deionized water and filled with the anodic solution.

Fig. 1. Atomic hydrogen permeation transients through a HY-130 steel membrane as a function of time for different applied cathodic potentials in a catholyte containing 1M Na2SO4, 0.4M NaCI, and 1M H_3BO_3 . The thickness of the membrane was $L = 0.15$ mm.

brane. Prepurified nitrogen was bubbled through both compartments in order to keep them free from oxygen contamination.

Results and Discussion

Atomic hydrogen permeation transients through a HY-130 steel membrane as a function of time for different applied potentials are shown in Fig. 1. The plateaux in Fig. 1 represent the j_{∞} values for different values of $E_{\rm c}$. The corresponding cathodic current density, i_c , was recorded for each value of E_c . As expected, the permeation current densities increased as the applied cathodic potentials and the resulting cathodic current densities were increased. Figure 2, a plot of j_x vs. i_c , is clearly linear.

Permeation transients through a palladium membrane were obtained as a function of applied cathodic current densities, as shown in Fig. 3. The plateaux in Fig. 3 represent j_x values for the associated set of i_c values. The permeation current density, j_{∞} , for palladium also increases linearly with the cathodic current density, as shown in Fig. 4.

Fig. 2. The steady-state hydrogen permeation current density \mathbf{i}_x **as a function of the cathodic current density (i<) through the HY-130 steel membrane.**

Fig. 3. Hydrogen permeation transients through a Pd membrane obtained at constant cathodic current densities. The thickness of the palladium membrane was $L = 25 \mu m$. The catholyte was 1N H₂SO₄.

Conclusion

The direct hydrogen entry mechanism^{1,2} was shown to be correct for HY-130 steel and Pd for small values of j_c . A simple expression (Eq. 9) was developed that showed that the hydrogen permeation current density, j_x , is directly proportional to the applied cathodic current density, i_c , and is independent of the thickness of the metal membrane. This expression was shown to be consistent with experimental data presented here and with data from previous workers.

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Fig. 4. The steady-state hydrogen permeation current density (i_a) **as a function of the cathodic current density (i,) through a palladium** membrane $(L = 25 \mu m)$.

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LIST OF SYMBOLS

-
- a F/RT , V^{-1}
*C*_o hydrogen hydrogen concentration directly beneath the cathode surface, mol cm
- $C_{\rm s}$ saturation value of $C_{\rm o}$, mol cm⁻³
- D hydrogen diffusivity, cm² s⁻¹
- $E_{\rm c}$ cathodic potential, V
-
- $i_{\rm c}$ cathodic current density, A cm⁻² $\,$ $i_{\rm o}$ exchange current density, A cm $^{-2}$
- C_s saturation value of C_o , mol cm⁻³
 D hydrogen diffusivity, cm² s⁻¹
 F_c cathodic potential, V
 F Faraday's constant, 96,487 C (eq)⁻¹
 i_o cathodic current density, A cm⁻²
 i_o exchange current densi $\frac{1}{2}$
steady-state hydrogen permeation current density,
 $\frac{1}{2}$ A cm
- k_1 absorption constant, mol (cm² s)⁻¹
 k_{-1} desorption constant, cm s⁻¹
-
- k_{-1} desorption constant, cm s
 k_3 hydrogen recombination r
 L membrane thickness, cm hydrogen recombination rate constant, mol (cm² s)⁻¹
- L membrane thickness, cm
- R gas constant, 8.3143 J(mol K)⁻¹ R gas constant, 8.
T temperature, K

Greek

- α_1, α_2 transfer coefficients, dimensionless η overpotential, V
- overpotential, V

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